

## Field induced resistivity anisotropy in SrRuO<sub>3</sub> films

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SrRuO<sub>3</sub> is an itinerant ferromagnet with orthorhombic structure and uniaxial magnetocrystalline anisotropy—features expected to yield resistivity anisotropy. Here we explore changes in the resistivity anisotropy of epitaxial SrRuO<sub>3</sub> films due to induced magnetization in the paramagnetic state by using the planar Hall effect. We find that the effect of the induced magnetization on the in-plane anisotropy is strongly angular dependent, and we provide a full description of this behavior at 160 K for induced magnetization in the (001) plane. © 2009 American Institute of Physics. [DOI: 10.1063/1.3072785]

The itinerant ferromagnet SrRuO<sub>3</sub> has attracted considerable experimental and theoretical effort for its intriguing properties.<sup>1–4</sup> Being almost cubic, thin films of SrRuO<sub>3</sub> exhibit rather small anisotropy which is difficult to characterize by comparing resistivity measurements taken on patterns with current flowing along different directions relative the crystallographic axes. Consequently, we have determined the zero field anisotropy of epitaxial films of this compound, by measuring the planar Hall effect (PHE),<sup>5</sup> a method which provides a good local measurement of the anisotropy with high accuracy. To explore the contribution of the magnetization  $\mathbf{M}$  to the anisotropy, we examine how changing the magnitude and orientation of the magnetization affects the resistivity anisotropy. We find that the anisotropy is proportional to  $M^2$  for any angle in the (001) plane; however, the proportionality coefficient is strongly angular dependent.

Our samples are epitaxial films of SrRuO<sub>3</sub> grown on slightly miscut (2°) SrTiO<sub>3</sub>. The films are orthorhombic ( $a = 5.53$  Å,  $b = 5.57$  Å, and  $c = 7.85$  Å) (Ref. 6) and their Curie temperature is  $\sim 150$  K. The films grow with the in-plane  $c$  axis perpendicular to the miscut, and with  $a$  and  $b$  axes at 45° relative to the film plane. The films exhibit uniaxial magnetocrystalline anisotropy with the easy axis along the  $b$  axis at  $T \geq T_c$ .<sup>6</sup>

The data presented here are from a 27 nm thick film with resistivity ratio of  $\sim 12$ . The measurements were done at 160 K on a pattern with current direction along  $[1\bar{1}1]$  for which the PHE attains its maximal value.<sup>5</sup>

Figures 1(a) and 1(b) show the transverse signal [(a) symmetric and (b) antisymmetric] versus  $\theta$ , the angle between the applied magnetic field and the normal to the film. Positive  $\theta$  corresponds to an anticlockwise rotation [Fig. 2(b), inset]. The PHE is the symmetric part of the transverse signal,<sup>7–9</sup> and in our configuration  $\rho_{\text{PHE}} = \rho_{[1\bar{1}0]} - \rho_{[001]}$ , where  $\rho_{[1\bar{1}0]}$  is the longitudinal resistivity along the  $[1\bar{1}0]$  direction and  $\rho_{[001]}$  is the longitudinal resistivity along  $[001]$ . The antisymmetric part ( $\rho_{\text{AS}}$ ) consists of the ordinary Hall effect

(OHE) and the extraordinary Hall effect (EHE).<sup>10</sup>

Subtracting the OHE contribution from  $\rho_{\text{AS}}$  yields the EHE which we present in Fig. 2(a). As is well known, the EHE is proportional to  $M_{\perp}$ , the component of  $\mathbf{M}$  which is perpendicular to the plane of the film. However, the magnetocrystalline anisotropy of SrRuO<sub>3</sub> allows us to determine the full vector  $\mathbf{M}$ . As the easy axis is at 45° to the film normal, a field  $\mathbf{H}$ , applied in the (001) plane at angle  $\theta$  relative to the normal of the sample's plane, creates magnetization  $\mathbf{M}$  at angle  $\alpha$  [Fig. 2(b), inset], whereas the same field at angle  $90 - \theta$  creates  $\mathbf{M}$  with the same magnitude but at angle  $90 - \alpha$ . Consequently, by measuring the  $\rho_{\text{EHE}}$  at angle  $\theta$  [ $\rho_{\text{EHE}}(\theta)$ ] and  $90 - \theta$  [ $\rho_{\text{EHE}}(90 - \theta)$ ], we obtain  $\rho_{\text{EHE}}(\theta) \propto M_{\perp}(\theta)$  and  $\rho_{\text{EHE}}(90 - \theta) \propto M_{\parallel}(\theta)$ , where  $M_{\parallel}$  and  $M_{\perp}$  are the in-plane and perpendicular components of  $\mathbf{M}$ , respectively. Thus we obtain a full description of the magnetization vector.<sup>11</sup>

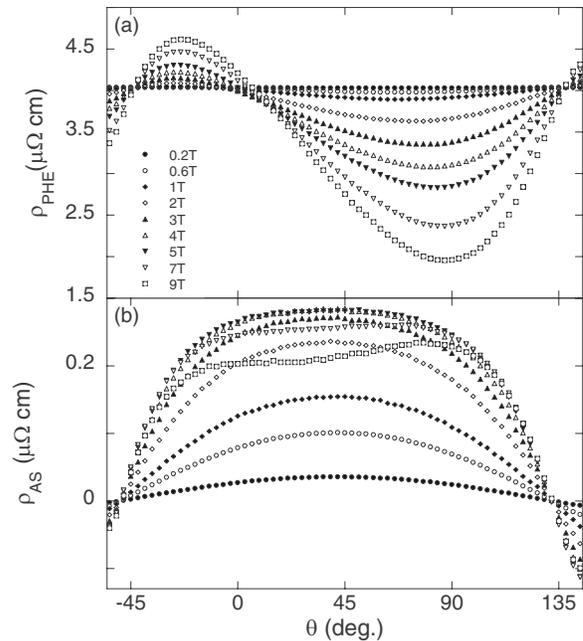


FIG. 1. (a)  $\rho_{\text{PHE}}$  and (b)  $\rho_{\text{AS}}$  vs  $\theta$ , the angle between the applied field and the film normal.

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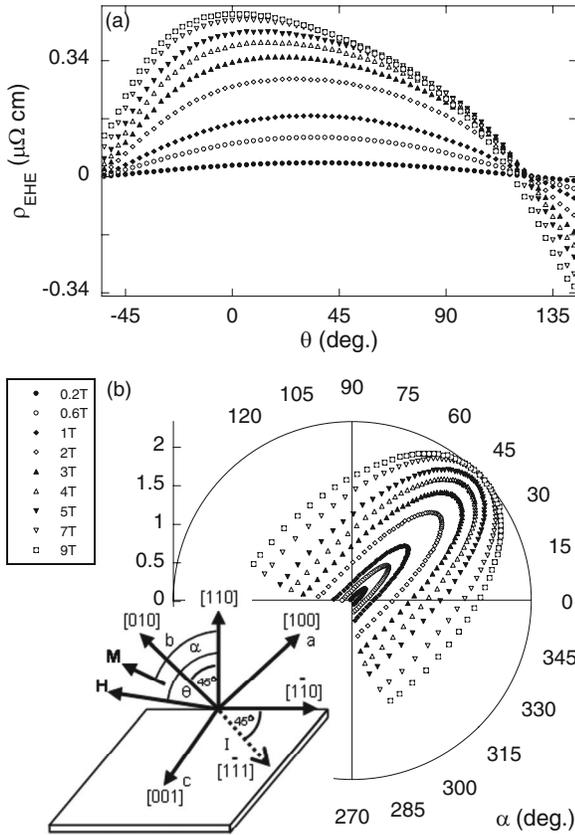


FIG. 2. (a) The EHE part of the antisymmetric signal vs  $\theta$  the angle between the applied field and the film normal. (b) The magnitude (arbitrary units) and direction of the magnetization obtained for rotating different fields in the (001) plane. Inset: Crystallographic directions of the film and definitions of the angles  $\theta$  and  $\alpha$ .

Figure 2(b) shows the vector of magnetization in arbitrary units obtained for some of the fields we used in our measurements. In this polar graph we see that the magnetization has its maximal value when it is aligned along the easy axis and its minimal value when it lies along the hard axis. The data indicate that with  $H=1$  T the induced magnetization along the easy axis is larger than the induced magnetization along the hard axis by a factor of  $\sim 5.5$ , in good agreement with a previous report.<sup>12</sup>

Figure 3 shows the dependence of  $\rho_{\text{PHE}}$  on  $m^2$  where  $\mathbf{m}$  is defined as  $\mathbf{M}/|\mathbf{M}|$  and  $\tilde{\mathbf{M}}$  is the magnetization obtained with a field of 1 T applied along the easy axis. Based on the

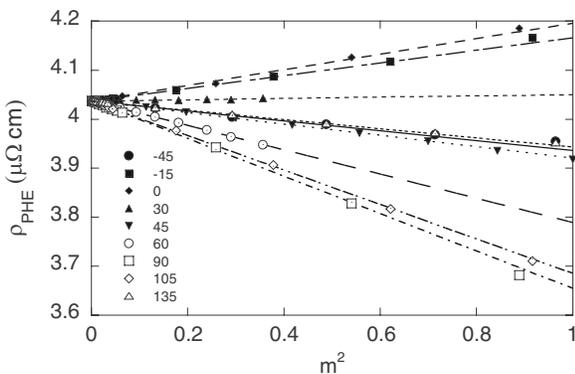


FIG. 3.  $\rho_{\text{PHE}}$  vs  $m^2$ . The lines are linear fits.

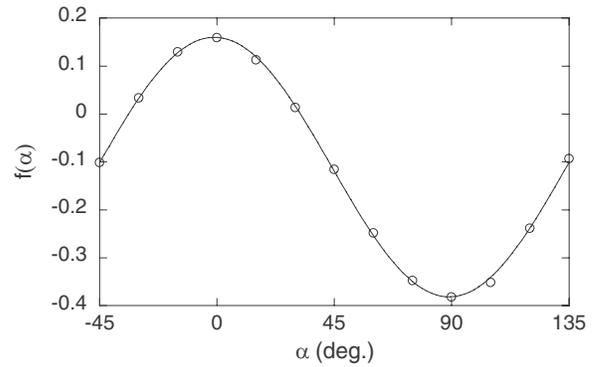


FIG. 4.  $f(\alpha)$  where  $\alpha$  is the angle between  $\mathbf{m}$  and the film normal. The solid line is a fit to Eq. (2).

construction of the vector of magnetization presented in Fig. 2, we can extract the change in  $\rho_{\text{PHE}}$  as a function of  $\mathbf{m}$  for a given orientation. We clearly see that

$$\rho_{\text{PHE}} = \rho_{\text{PHE}}^0 + f(\alpha)m^2, \tag{1}$$

where  $\rho_{\text{PHE}}^0$  is the PHE measured at zero field and  $f(\alpha)$  is the coefficient of  $m^2$  when  $\mathbf{m}$  is along  $\alpha$ . Figure 4 shows  $f(\alpha)$  for some values of  $\alpha$ . We see that there is an excellent fit of  $f(\alpha)$  with

$$f(\alpha) = a + b \sin^2 \alpha. \tag{2}$$

This means that one can write  $\rho_{\text{PHE}}$  as

$$\rho_{\text{PHE}} = \rho_{\text{PHE}}^0 + am^2 + bm_{\parallel}^2, \tag{3}$$

where  $a=0.16$  and  $b=-0.54$ .

The dependence we have found indicates that there are two effects of the induced magnetization on the resistivity anisotropy. First, there is a contribution independent of the direction of  $\mathbf{M}$  with a positive coefficient  $a$ , which means that the anisotropy increases with increasing magnetization. Second, there is another contribution sensitive to the in-plane projection of the magnetization with a negative coefficient  $b$ , which means that the anisotropy decreases with increasing magnetization. The extremal points of  $f(\alpha)$  are at  $\alpha=0^\circ$  and  $90^\circ$ . When  $\alpha=0^\circ$  the induced magnetization is normal to  $J$  along both  $[1\bar{1}0]$  and  $[001]$ . At this angle the anisotropy contribution is minimal and  $f(\alpha)$  attains its maximal positive value. When  $\alpha=90^\circ$ , the magnetization is parallel to  $[1\bar{1}0]$  and perpendicular to  $[001]$  and therefore its anisotropic contribution is maximal and  $f(\alpha)$  attains its maximum negative value.

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